

# SANS from Polymers

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## Outline of Talk

Introduction to Polymers and SANS

Example Experiments

Chain conformation

$R_g$  in thin films

$R_g$  of unusual architecture polymers

Thermodynamics, phase diagrams and interactions

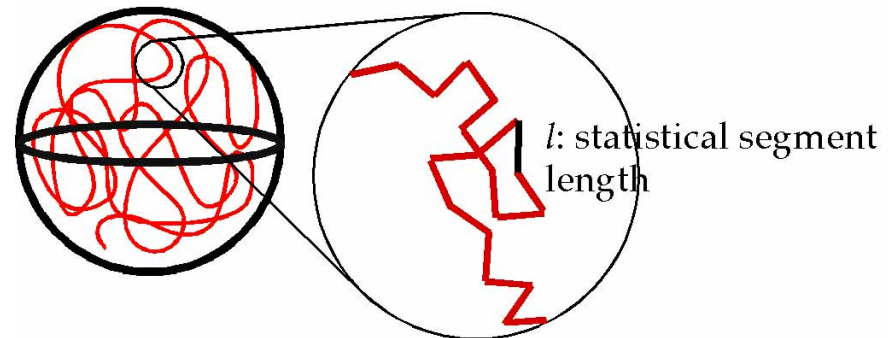
Polymer Blends

Block copolymers

Summary

# Chain Conformation

## Chain Conformation



## For a Gaussian Chain

$$R_g^2 = Nl^2/6 \quad (\text{general scaling: } R_g \sim N^v)$$

### Other conditions

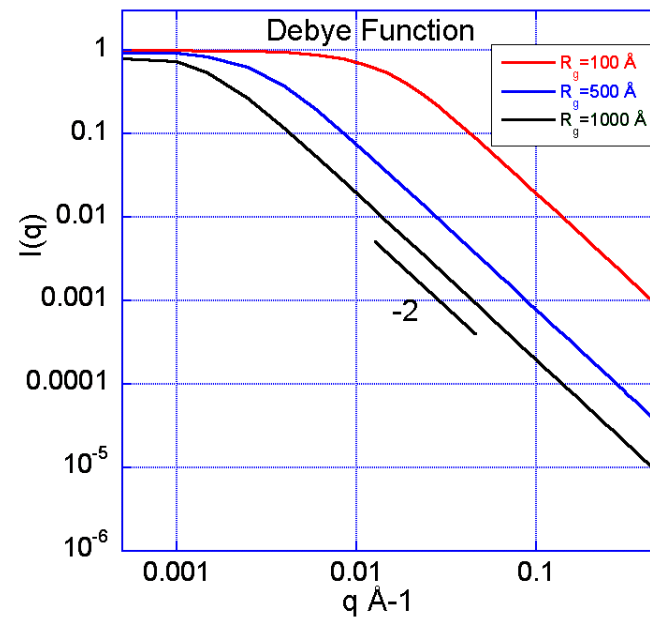
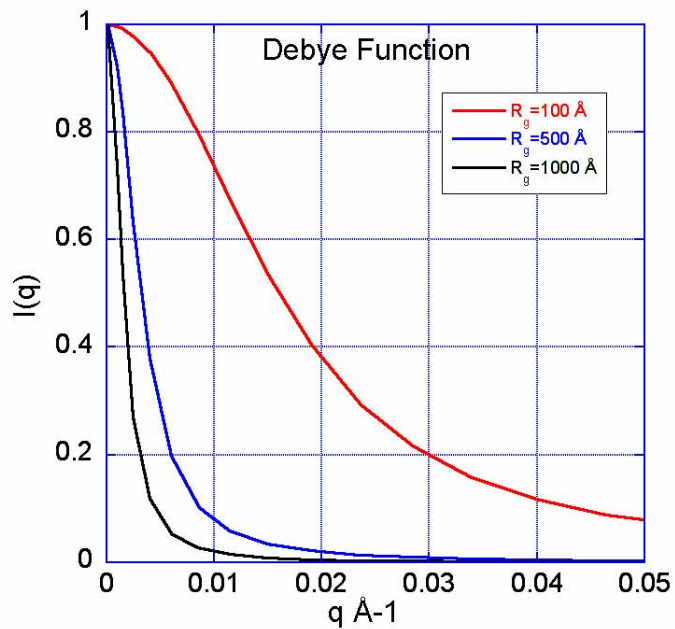
$$v = \begin{array}{l} 3/5 \text{ good solvent} \\ 1/2 \text{ theta solvent or melt} \\ 1/3 \text{ collapsed (constant density)} \end{array}$$

- $R_g$  controls many structural features in polymers
- Typical  $R_g$  for most polymers 50-500Å -perfect size scale for SANS

# Chain Conformation -cont.

Scattering from a Gaussian chain (Debye function):

$$P(q) = \frac{2}{x^2} (\exp(-x) - 1 + x) \quad x = R_g^2 q^2$$



# Chain Conformation -cont.

## Debye Function with a Zimm-Schulz molecular weight distribution

$$\langle P(q) \rangle_w = \frac{2}{\langle x \rangle_n^2} \left[ \langle x \rangle_n - 1 + \left( \frac{h}{h + \langle x \rangle_n} \right)^h \right]$$

where

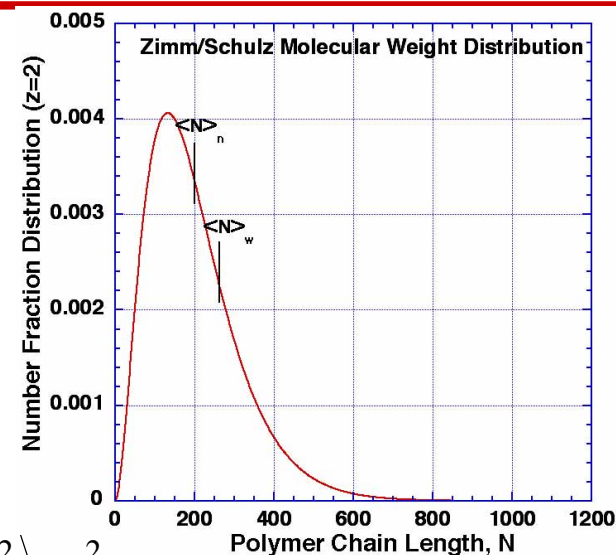
$$h = \frac{1}{\left[ \frac{\langle N \rangle_w}{\langle N \rangle_n} - 1 \right]}$$

$\langle N \rangle_n$  number average degree of polymerization

$\langle N \rangle_w$  weight average degree of polymerization

$$\langle x \rangle_n = \langle N \rangle_n l^2 q^2 / 6 = \langle R_g^2 \rangle_n q^2$$

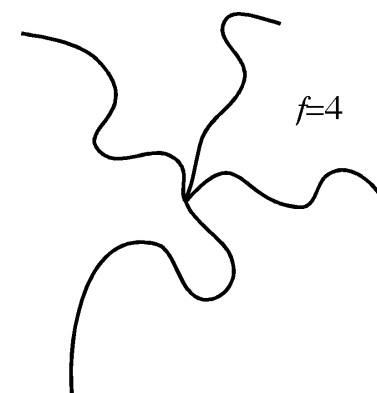
K. Mori, H. Hasegawa, T Hashimoto; *Polymer*, **1989**, 30, 1389



## Scattering function for a Gaussian star molecule

$$P(q) = \frac{2}{x^2} \left( \frac{x}{f} - \frac{(1 - \exp(-x))}{f} + (1 - \exp(-x)) \frac{2(f-1)}{2f} \right)$$

$$x = \frac{R_g^2 q^2}{f^g} \quad g = \frac{(3f-2)}{f^2} \quad f \text{ is the number of arms}$$



H. Benoit; *J. of Polymer Science*, **1953**, 11, 507-510

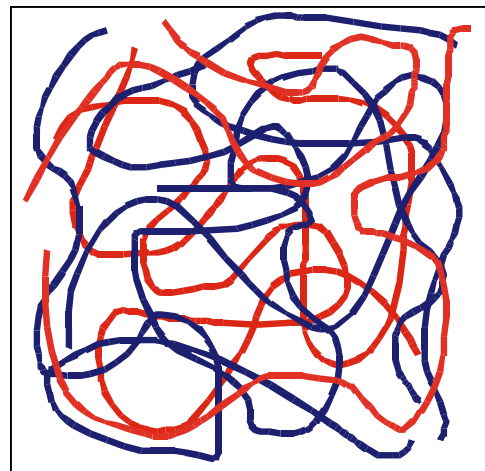
# Thermodynamics of Polymer Blends

## Flory-Huggins Free Energy of Mixing

$$\frac{\Delta f_{mix}}{kT} = \underbrace{\frac{\phi_a \ln \phi_a}{N_a} + \frac{\phi_b \ln \phi_b}{N_b}}_{\text{Entropy of mixing}} + \underbrace{\phi_a \phi_b \chi}_{\text{Enthalpy of mixing}}$$

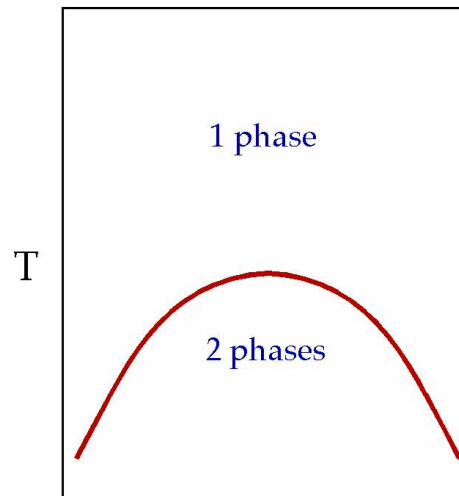
$\chi$ : Flory interaction parameter

$\chi \sim 1/T$



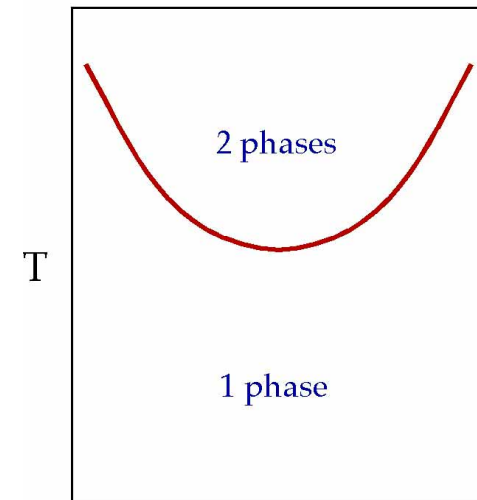
Miscible binary polymer blend

UCST Phase diagram



$\chi = A + B/T$  w/ B positive

LCST Phase diagram



$\chi = A + B/T$  w/ B negative



# Thermodynamics of Polymer Blends

## de Gennes Random Phase Approximation (RPA) for binary polymer blend

2 identical polymers (one labeled), no interactions ( $\chi=0$ )

$$I(q) = k_n \phi(1 - \phi)NP(q) \quad k_n \text{ neutron contrast factor}$$

2 polymers, different molecular weights, with interactions ( $\chi \neq 0$ )

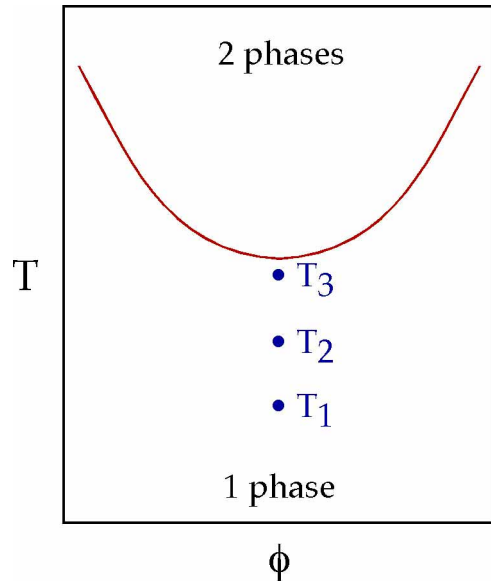
$$\frac{k_n}{I(q)} = \frac{1}{N_a \phi_a P_a(q)} + \frac{1}{N_b \phi_b P_b(q)} - 2\chi$$

q=0 limit

$$\lim_{q \rightarrow 0} \left[ \frac{k_n}{I(q)} \right] = \frac{1}{N_a \phi_a} + \frac{1}{N_b \phi_b} - 2\chi = \frac{\partial^2 \Delta f / kT}{\partial \phi^2}$$

Flory-Huggins  $\Delta f$   
Equation of the spinodal line

# Thermodynamics of Polymer Blends



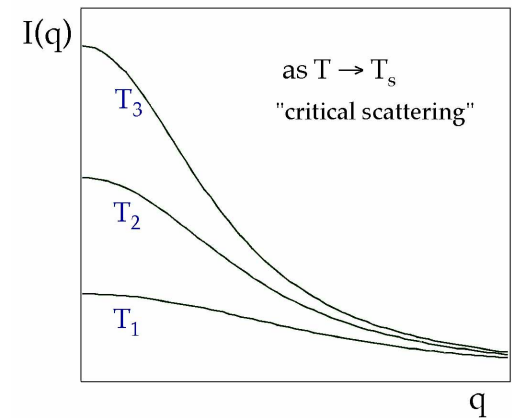
Measure SANS at  $T_1, T_2, T_3$

RPA eq.

$$\frac{k_n}{I(q)} = \frac{1}{N_a \phi_a P_a(q)} + \frac{1}{N_b \phi_b P_b(q)} - 2\chi$$

Expand  $P(q)$  in small  $q$  limit and RPA form reduces to O-Z form (Lorentzian)

RPA scattering (~Lorentzian shape)



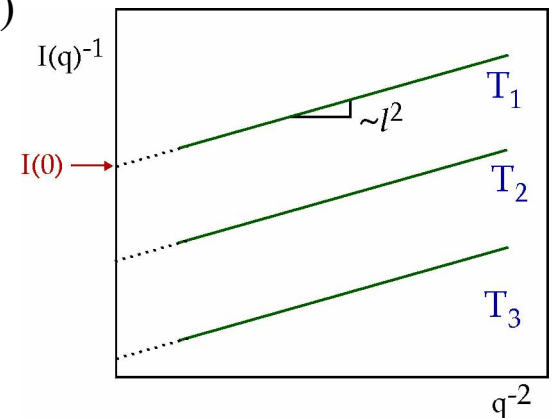
$$I(q) = \frac{K I(0)}{1 + \xi^2 q^2}$$

with

$$\xi^2 = \frac{\bar{l}^2}{36} [\phi_a \phi_b (\chi_s - \chi)]^{-1} \quad \text{correlation length}$$

$$\bar{l}^2 = \phi_a \phi_b \left( \frac{l_a^2}{\phi_a} + \frac{l_b^2}{\phi_b} \right) \quad \text{average statistical step length}$$

$$\chi_s = \frac{1}{2} \left( \frac{1}{N_a \phi_a} + \frac{1}{N_b \phi_b} \right) \quad \text{value of } \chi \text{ parameter at spinodal}$$

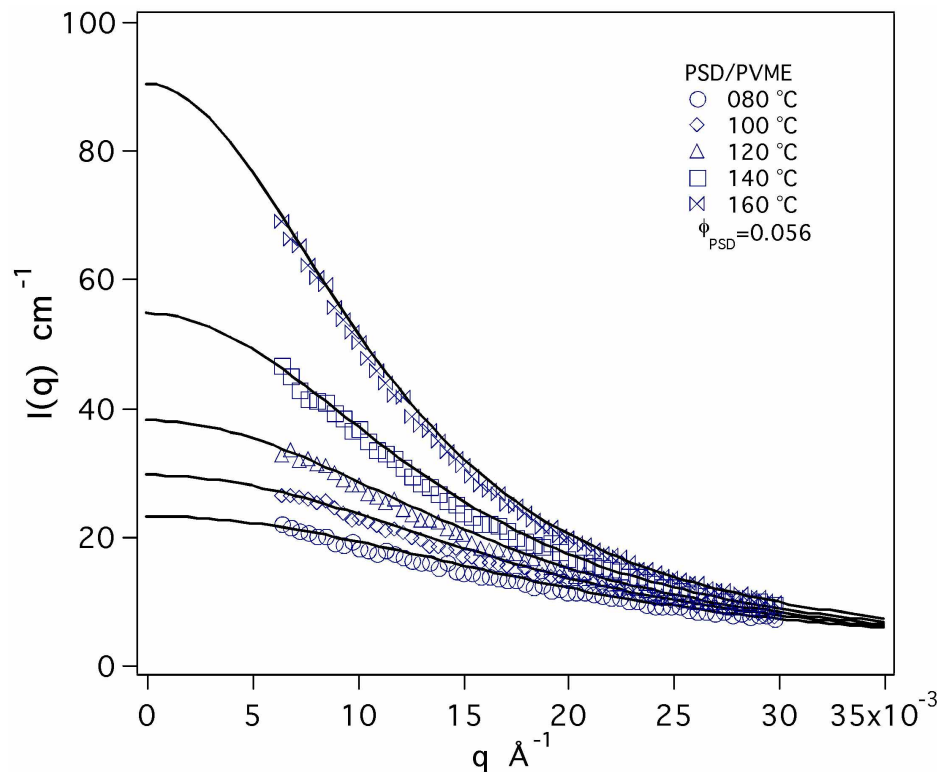


Linearized scattering (O-Z or Zimm plot)

# Thermodynamics of Polymer Blends



## SANS data from a polymer blend



## Fit the data with RPA equation

$$\frac{k_n}{I(q)} = \frac{1}{N_a \phi_a P_a(q)} + \frac{1}{N_b \phi_b P_b(q)} - 2\chi$$

### Known parameters

- $N_i$  chain length
- $\phi_2$  composition

### Fitting parameters:

- $\bar{l}$  average segment length
- $\chi$  Flory interaction parameter

### Calculated parameters:

- $\xi$  concentration fluctuation correlation length
- $I(0)$  zero angle scattering



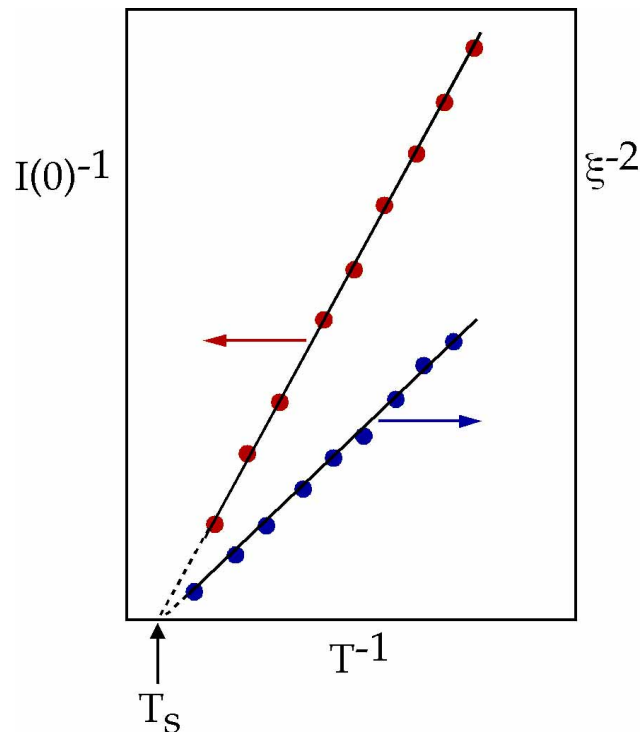
# Thermodynamics of Polymer Blends



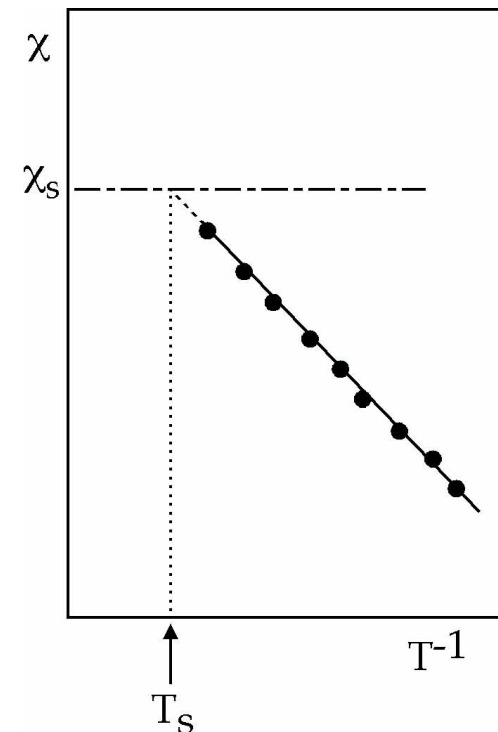
## Determine Phase Diagram

Mean-field scaling of  $\xi$  and  $I(0)$

$$I(0)^{-1} \rightarrow 0 \text{ as } T \rightarrow T_s$$
$$\xi^{-2} \rightarrow 0 \text{ as } T \rightarrow T_s$$



$\chi \rightarrow \chi_s$  as  $T \rightarrow T_s$

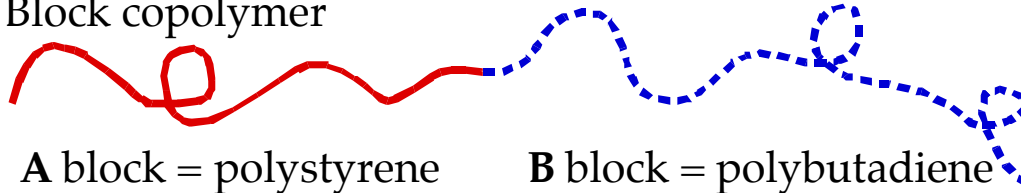


# Thermodynamics of Block Copolymers



## Block Copolymer Morphology

A/B Block copolymer



A block = polystyrene

B block = polybutadiene

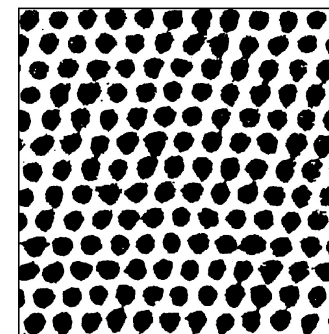
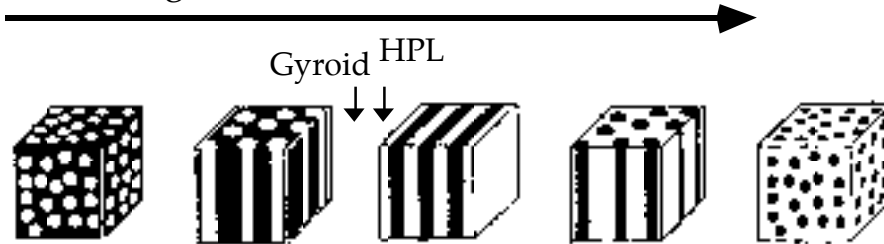
A & B are thermodynamically incompatible

A-B junction between blocks leads to MICROPHASE SEPARATION

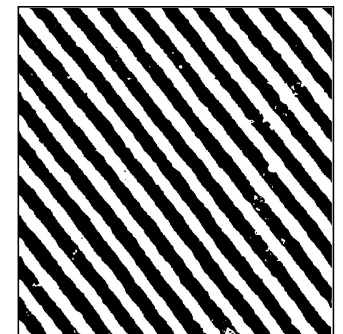
Microphase separation: phase separation constrained by size of blocks

## Microphase separation morphology

Increasing A block content (f)



End-on view of hexagonally packed cylinders



Lamellar morphology

1000 Å

TEM Micrographs

# Thermodynamics of Block Copolymers



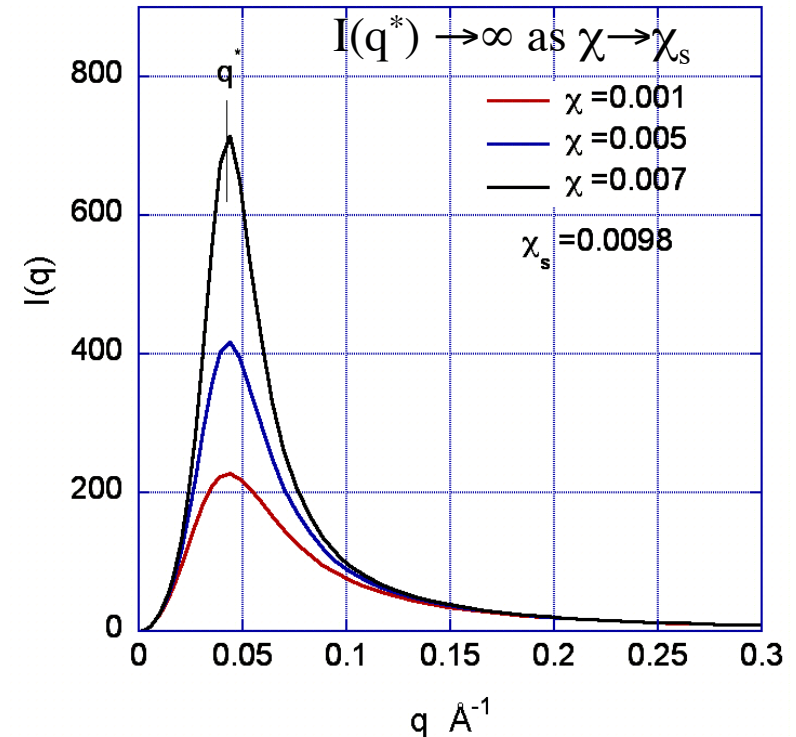
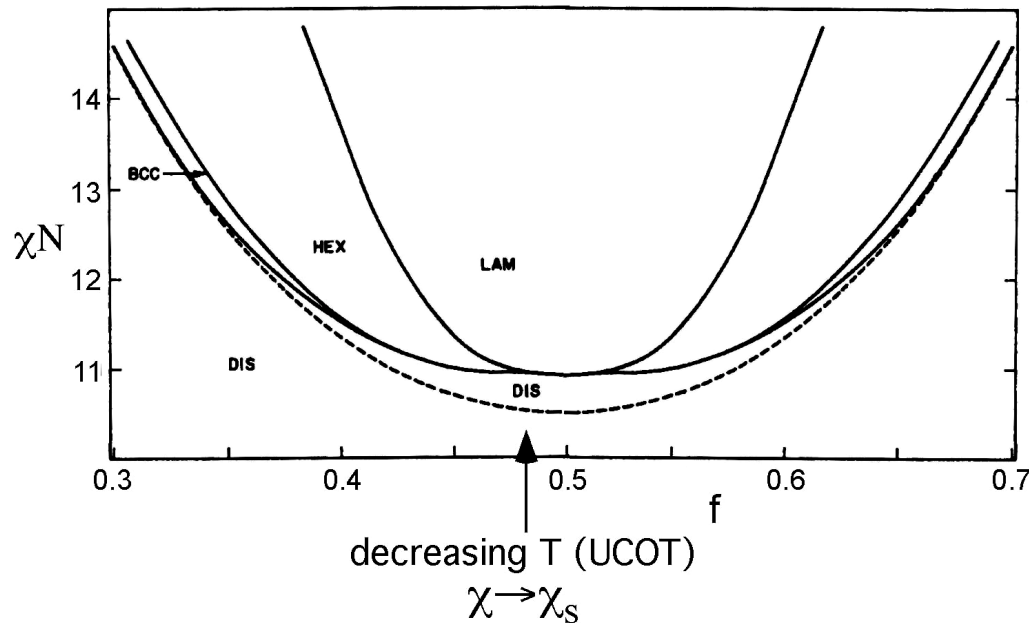
## Leibler Scattering function for a diblock copolymer (single phase region)

$$\frac{k_n}{I(q)} = \frac{(P_{aa} + P_{bb} + P_{ab})}{(P_{aa}P_{bb} - P_{ab}^2) - 2\chi}$$

with

$$P(f, x) = \frac{2}{x^2} (fx + \exp(-fx) - 1)$$

$$x = R_g^2 q^2 \quad f = \frac{N_a}{N_a + N_b}$$



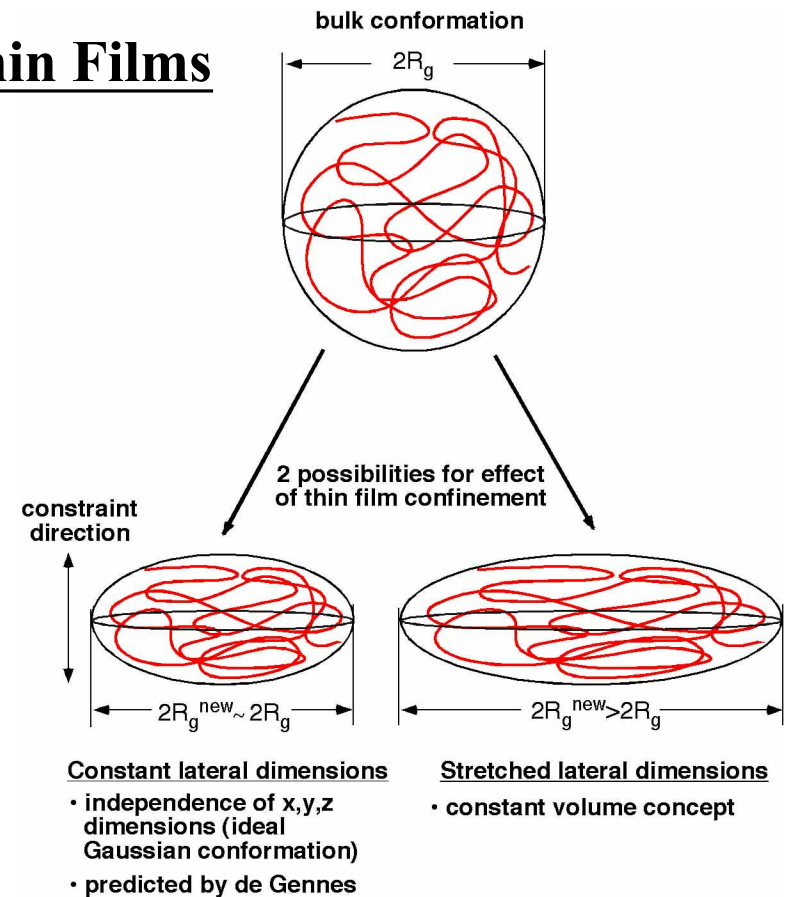
L. Leibler; *Macromolecules*, **1980**, 13, 1602-1617

G.H. Fredrickson, E.Helfand, *J. Chem. Phys.*, **1987**, 8791), 697

# Examples -Chain Conformation

## Polymer Chain Conformation in Ultra thin Films

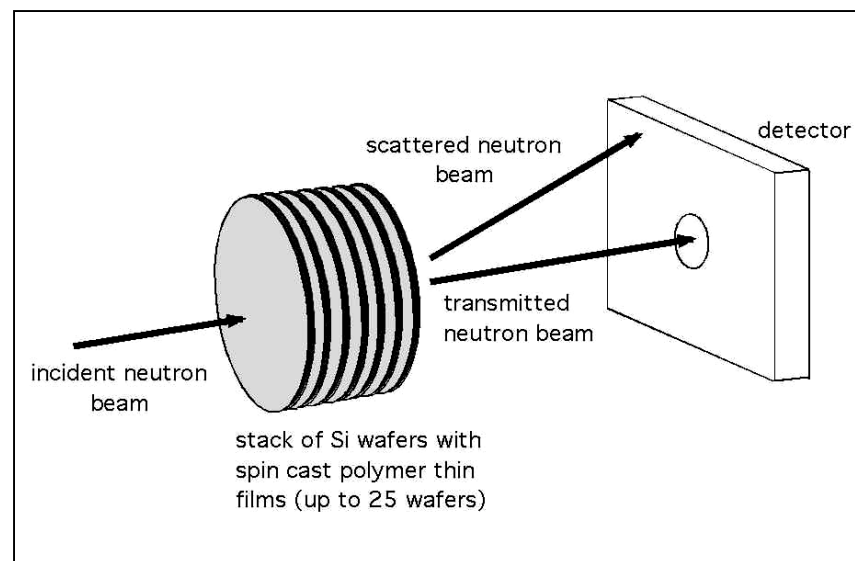
- **Objective:** Characterize polymer chain conformation in ultra thin films (film thickness  $< 2R_g$ )
  - **Polymer chain conformation in ultra thin films is a fundamental physical question which remains open**
  - **Development of the technique of small angle neutron scattering (SANS) for the study of thin polymer films**
  - **Polymer conformation plays a critical role in film wetting, adhesion and coatings**



# Examples -Chain Conformation

- **Approach**

- Use SANS to measure the radius of gyration ( $R_g$ ) of the polymer chain in ultra thin films on Si wafers (neutron transmission of Si  $\sim 0.999+$ )
- Mixtures of deuterated polystyrene and normal polystyrene. Deuterium labeling provides neutron contrast. 25 d-PS/75 h-PS composition.



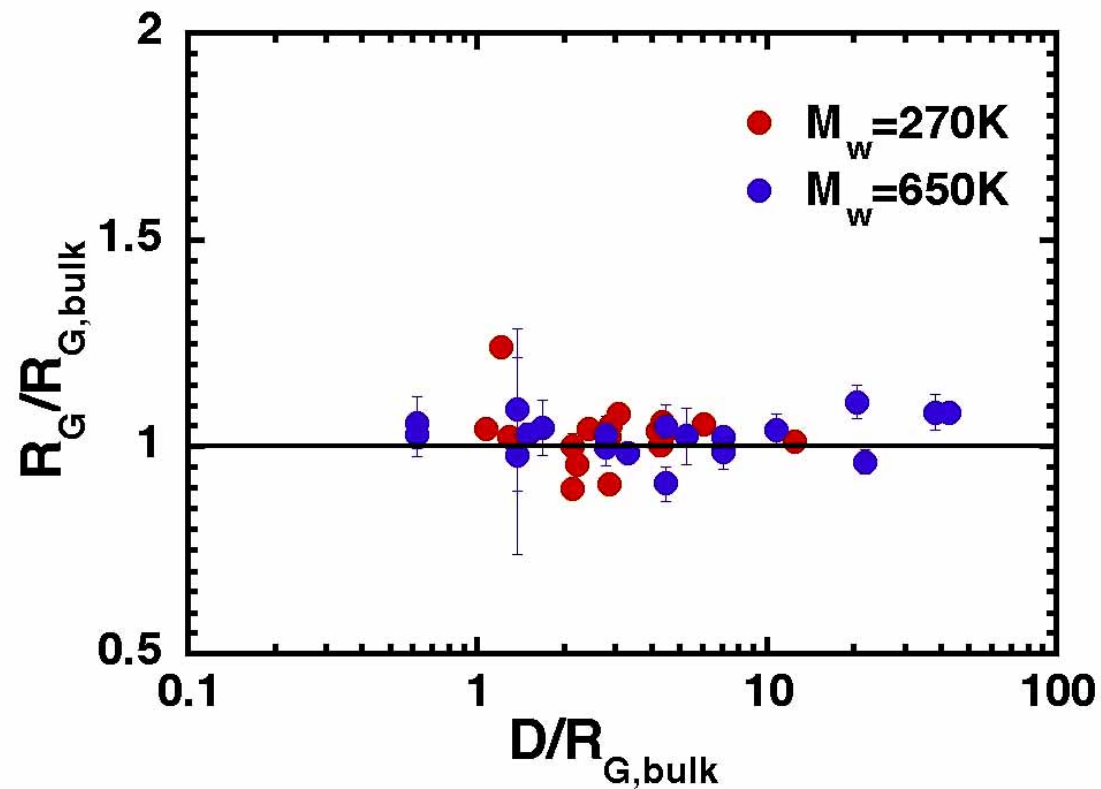
Thin films are spun on 1" Si wafers (0.75" neutron beam diameter)

For stacks of 25 wafers with a film thickness of 10nm total polymer sample is  $\sim 0.2$  mg

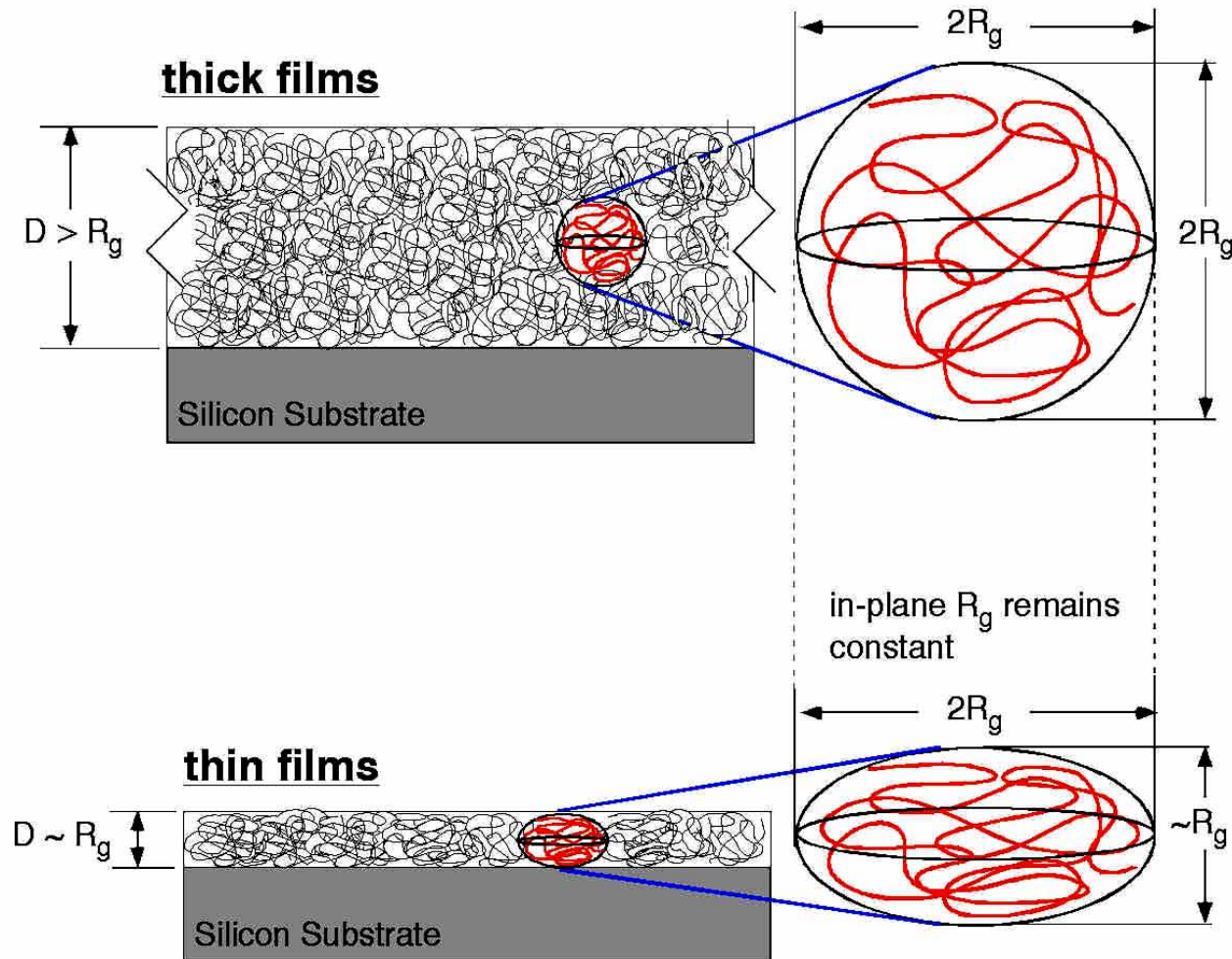
# Examples -Chain Conformation

## Results:

$R_g$  in plane of film remains unchanged down to film thickness  $\sim 1/2 R_g$ !



# Examples -Chain Conformation



## Implications

- Chain conformation in x,y,z directions are independent
- Higher segment density inside coil volume for thin films ( $D < 2R_g$ )
- Less entanglements per chain in thin films
- Different chain dynamics in thin films

$R_g$  normal to film decreased by thickness constraint

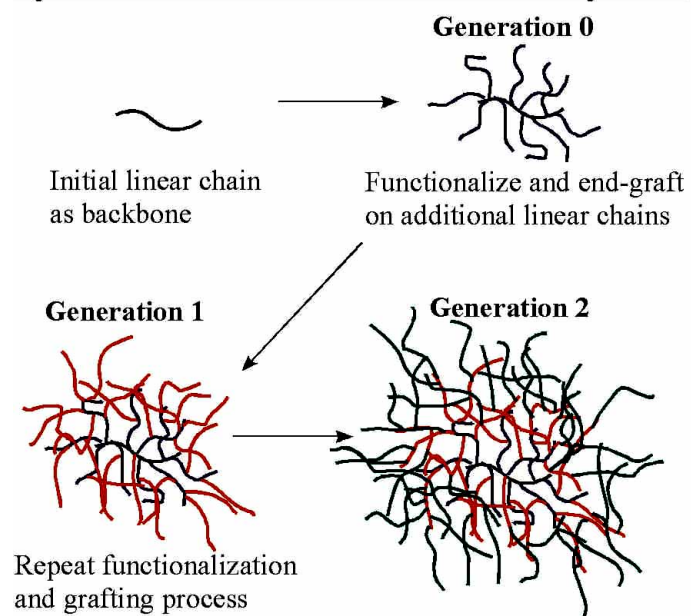


# Examples -Chain Conformation

## Characterization of Arborescent Graft Polymers

- **Objective:** Characterize the behavior of arborescent graft polymers in solutions and in blends with linear polymers
- Arborescent graft polymers are new molecules with an unusual chain architecture. The goal is to use small angle neutron scattering to measure the size and shape in solutions and blends.
- The characterization of the size, shape and density profile of arborescent graft polymers will provide insight useful for tailoring them to meet end use requirements as unimolecular micelles, drug delivery vehicles and flow modifiers.

### Synthesis of Arborescent Graft Polymers



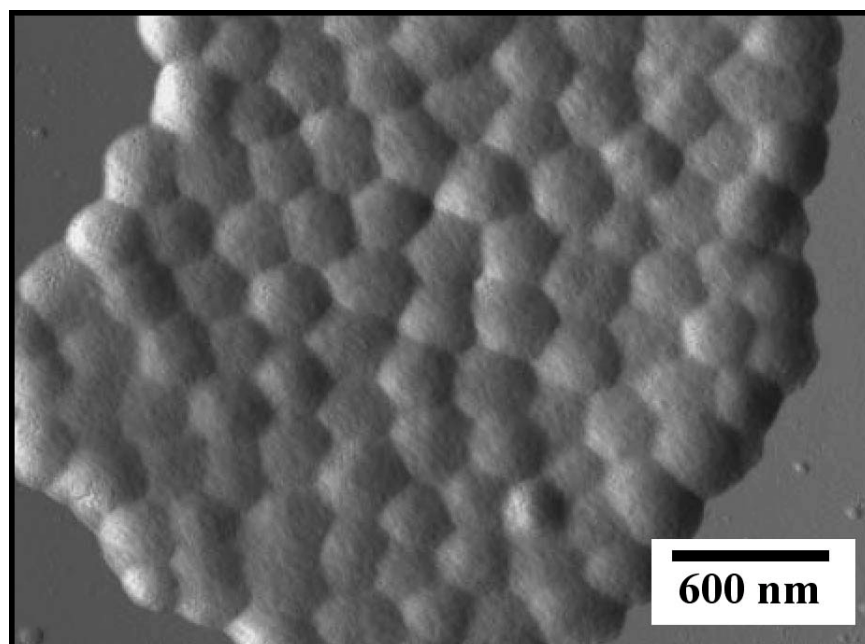
#### Notes

- very high grafting density (~0.15-0.2 grafts/monomer)
- Molecular weight increases exponentially



# Examples -Chain Conformation

## AFM Image

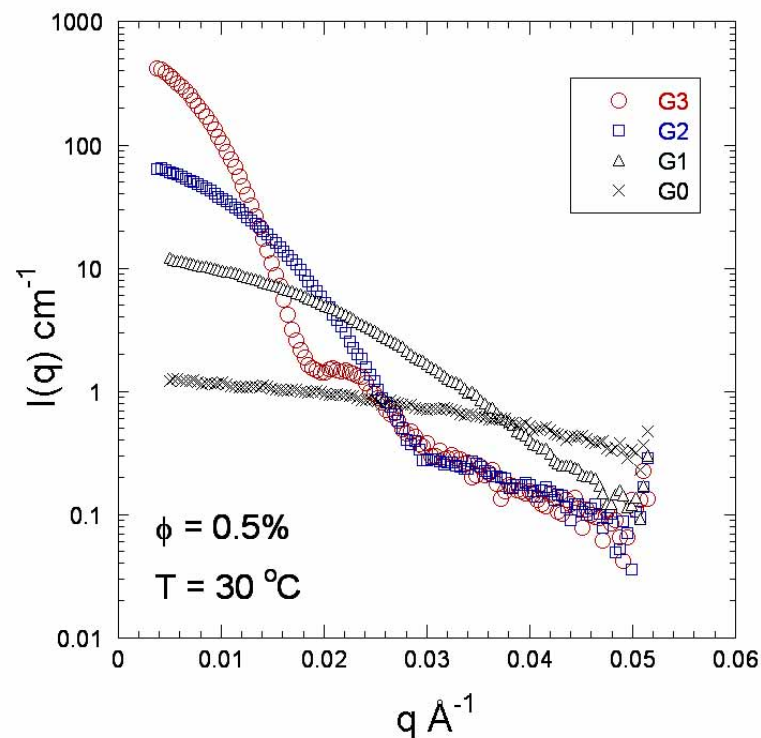


AFM micrograph of a film of 3<sup>rd</sup> generation AGP molecules synthesized from 30k  $M_w$  PS branches.

## • Approach:

- Use small angle neutron scattering to measure  $R_g$  and  $\rho(r)$  in solutions and blends.
- Deuterated solvents and linear polymers are used to provide neutron contrast.

### 5k Series in d-cyclohexane



# Examples -Chain Conformation

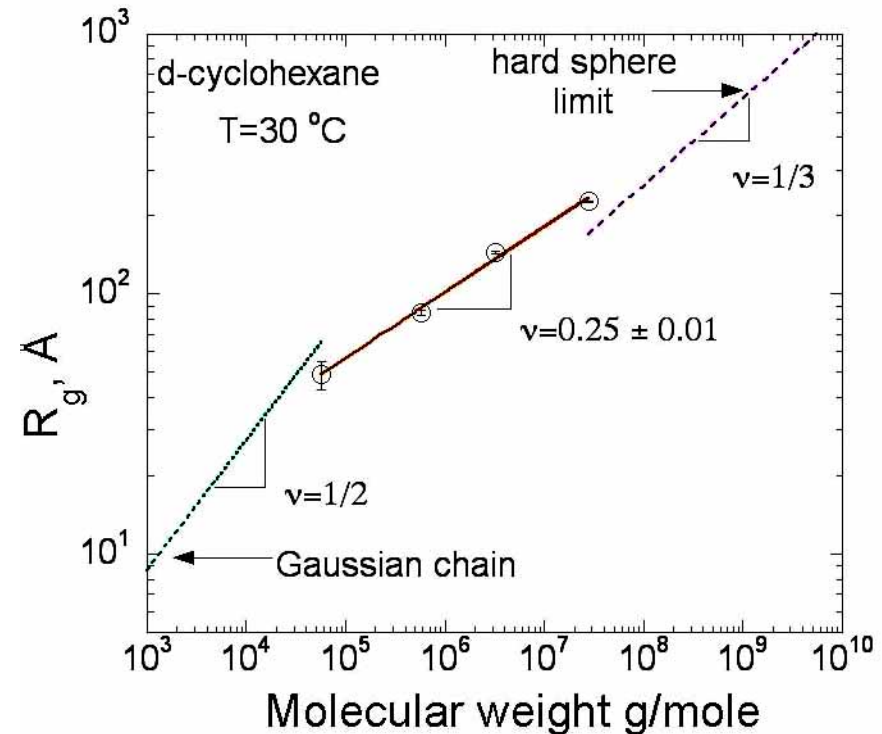
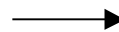
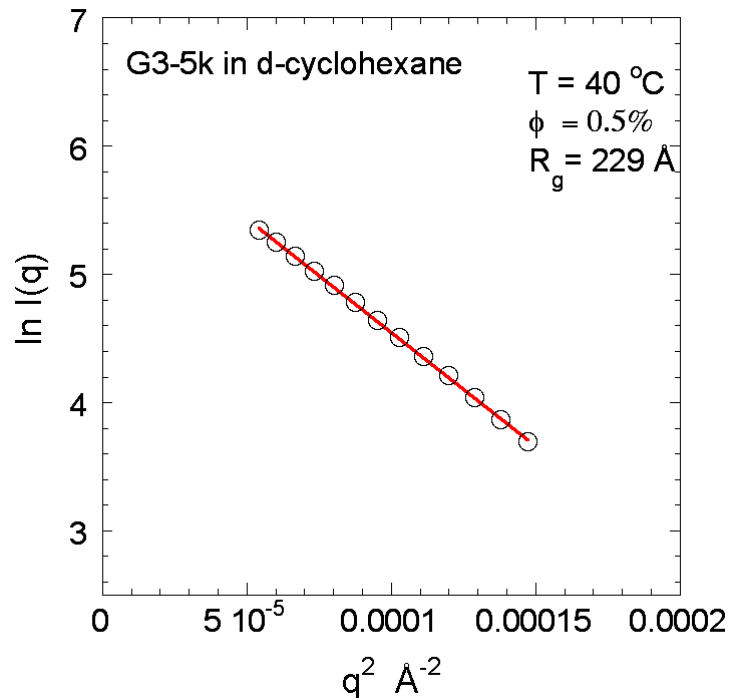
## Dilute Solution: $-R_g$ of AG polymer from Guinier Analysis

$$I(q) = k_n \phi(1 - \phi) NP(q)$$

In the small  $q$  limit,  $P(q) \rightarrow \exp(-R_g^2 q^2/3)$

$$I(q) = k_n \phi(1 - \phi) \exp\left(-\frac{R_g^2 q^2}{3}\right)$$

Plot  $\ln I(q)$  vs.  $q^2$

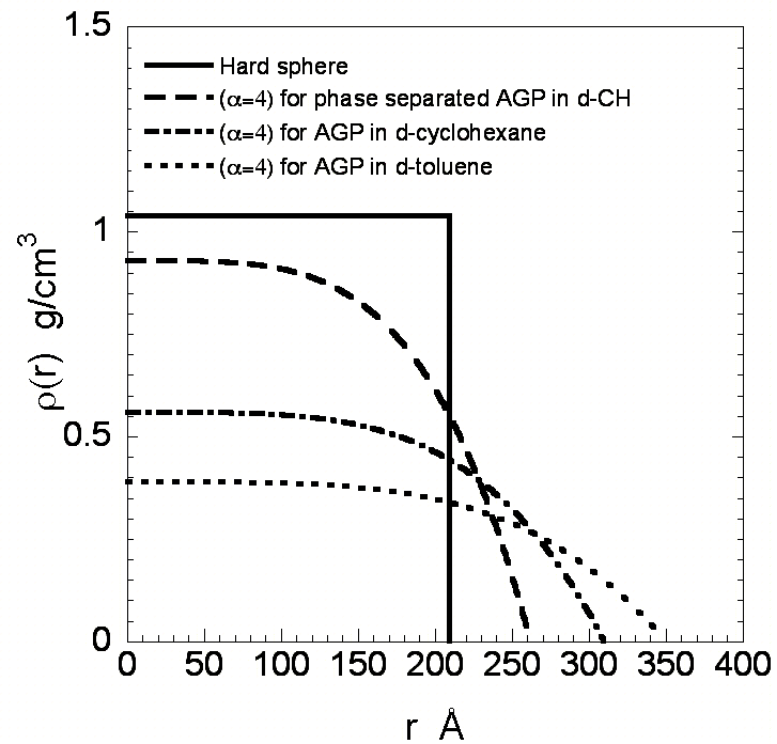
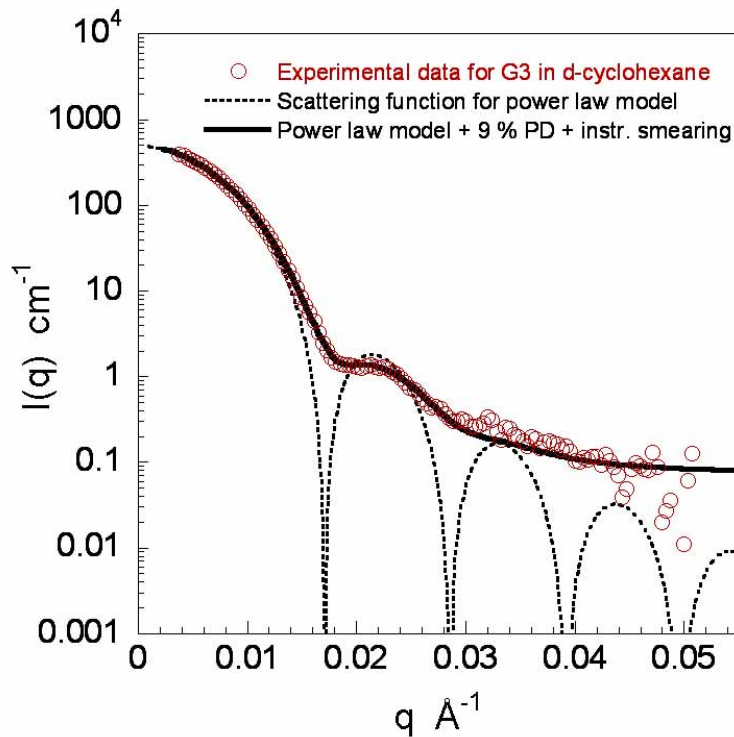


**Scaling of  $R_g$  with  $M_w$  of the form  $R_g \sim M_w^\nu$  with  $\nu=0.25$  which indicates that arborescent graft polymers become more dense with increasing size (molecular weight).**

# Examples -Chain Conformation

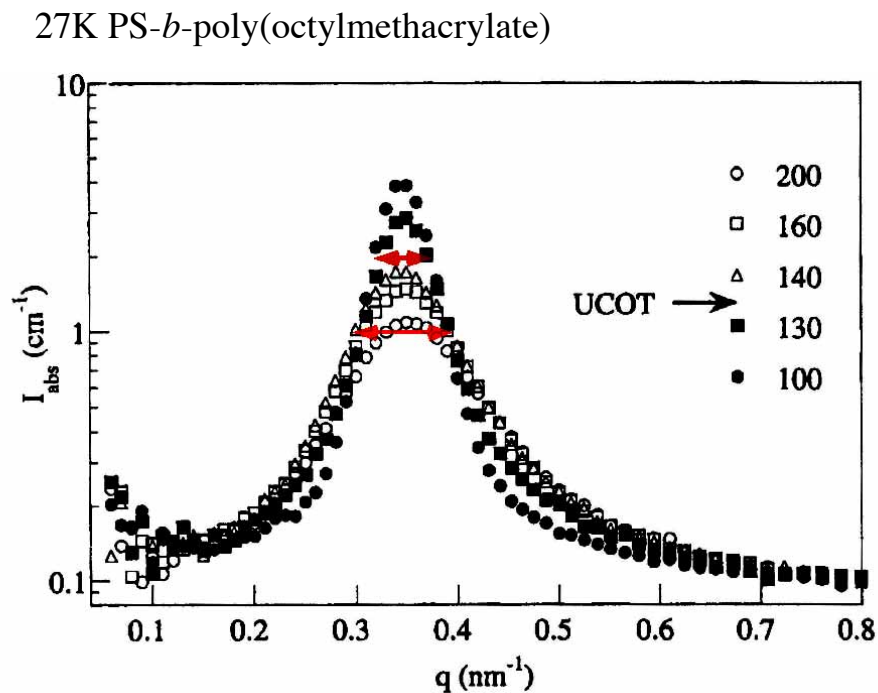
Model density profile  $\rho(r) = 1 - \left(\frac{r}{R_{\max}}\right)^\alpha$

Calculate scattering (with instrumental and size distribution smearing)  $I(q) = \left[ \int \rho(r) \frac{\sin(qr)}{qr} r^2 dr \right]$



# Examples -Block Copolymers

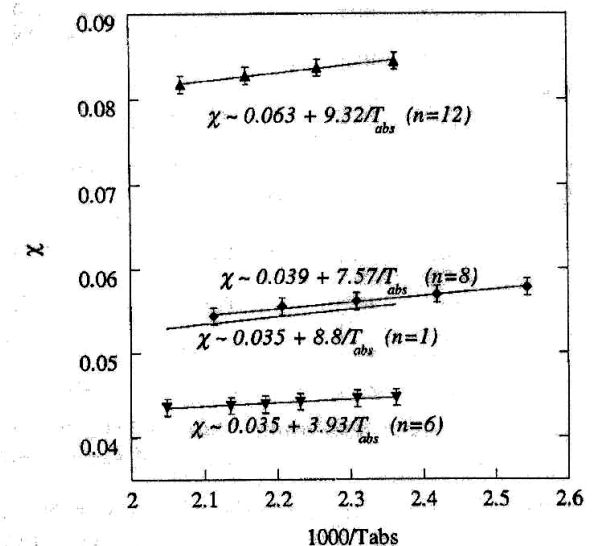
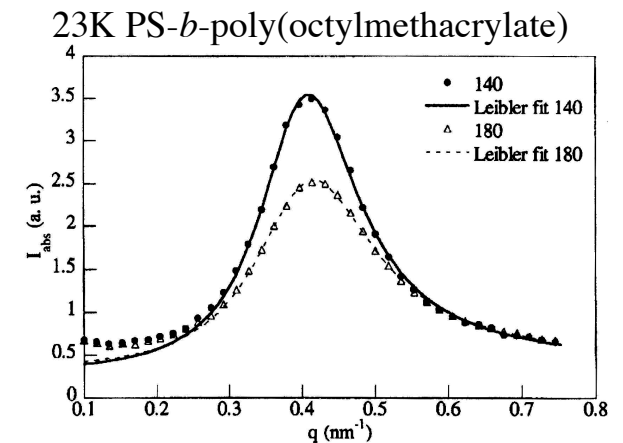
## Phase mixing on heating (UCOT)



$I(q^*) \rightarrow \infty$  as  $\chi \rightarrow \chi_s$  (not reliable)

FWHM broadens in single phase

Fit to Leibler scattering function with  $\chi$ ,  $l$  as fit parameters

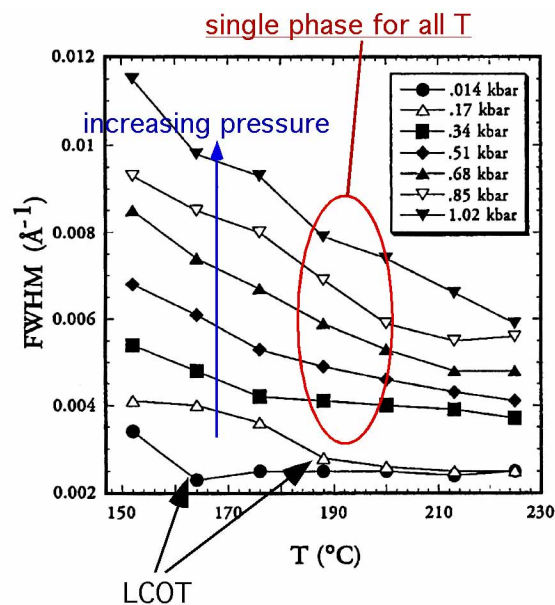
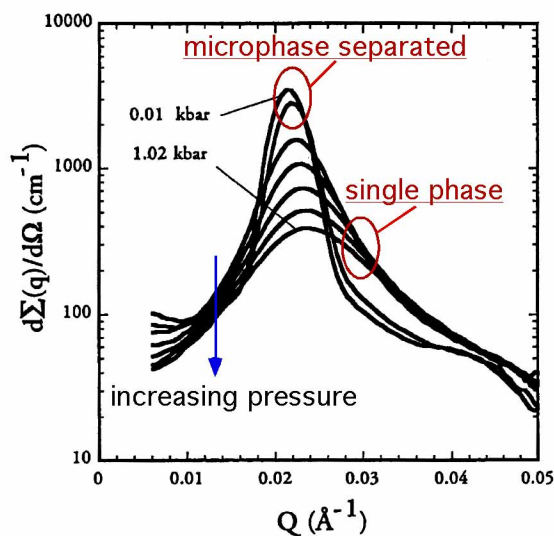


# Examples -Block Copolymers

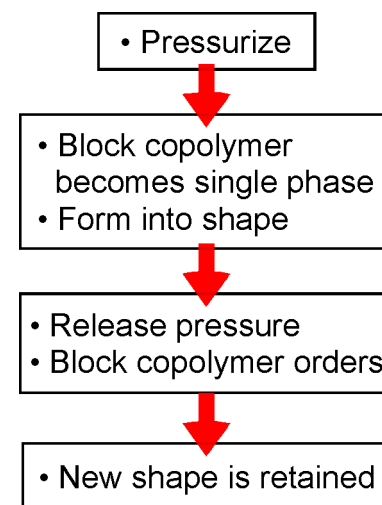
## Pressure Effects on the Ordering Transition in Block Copolymers

- LCOT block copolymer disorders with increasing pressure
- Allows for the intelligent design of a baroplastic elastomer

85K d-PS-*b*-poly(*n*-butyl methacrylate)



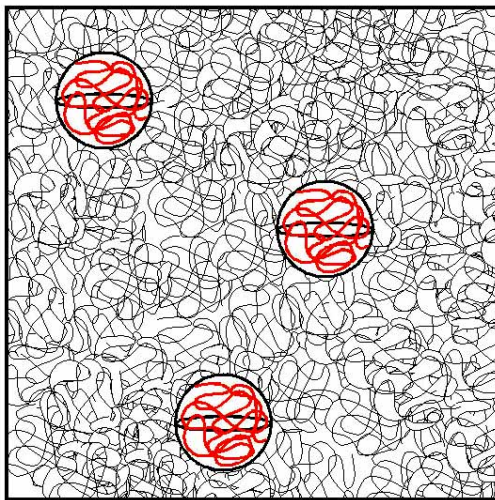
### Baroplastic Elastomer



# Examples - Polymer Blends

## Chain Conformation in Dilute Polymer Blends

Polymer Blend with  
one species dilute



### Questions

- $R_g \sim f(T)$
- Collapse transition in blends

Simulations of  $R_g \sim f(T)$   
in dilute blends

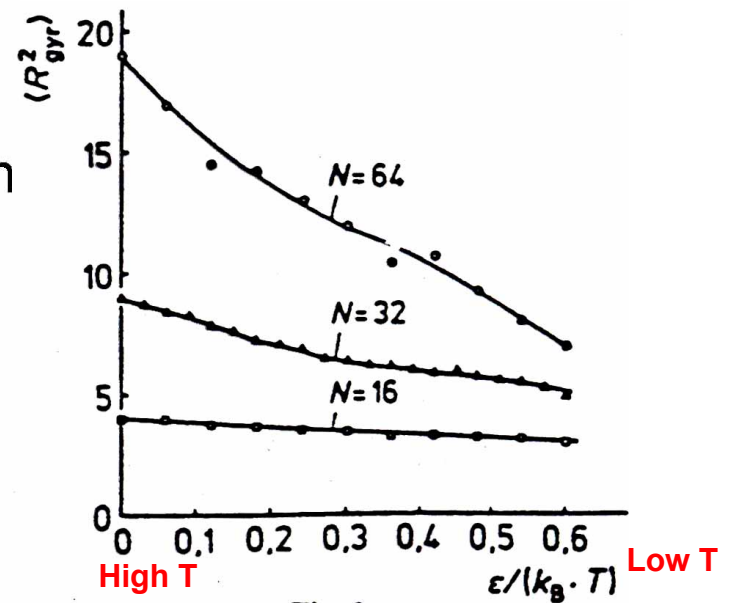


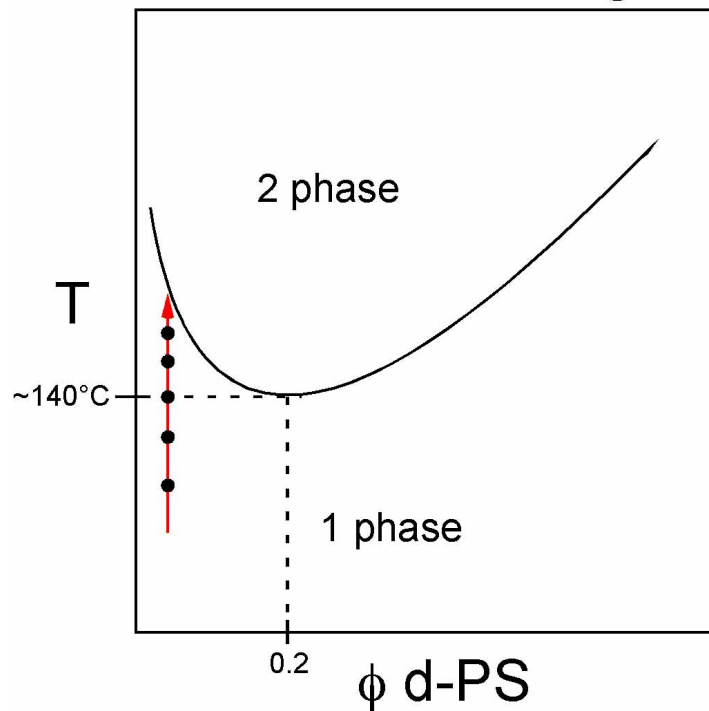
Fig. 2.

Sariban and Binder, *Mackromol. Chemie*,  
1988, 189, 2357



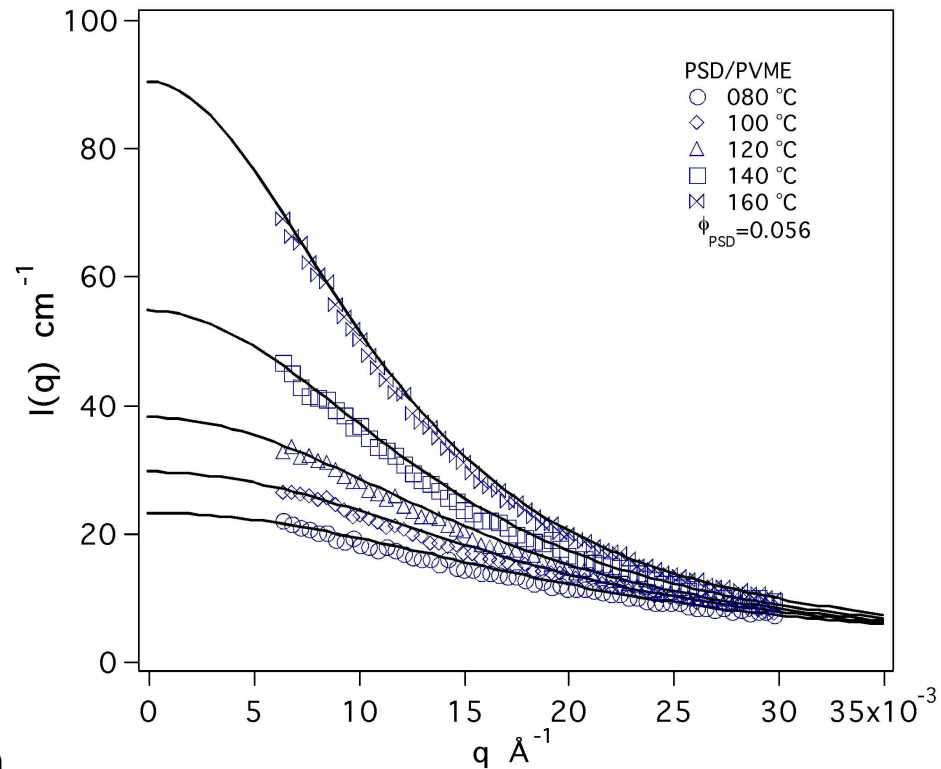
# Examples - Polymer Blends

**d-PS/PVME Phase Diagram**



Fit the data with the RPA equation

$$\frac{k_n}{I(q)} = \frac{1}{N_a \phi_a P_a(q)} + \frac{1}{N_b \phi_b P_b(q)} - 2\chi$$



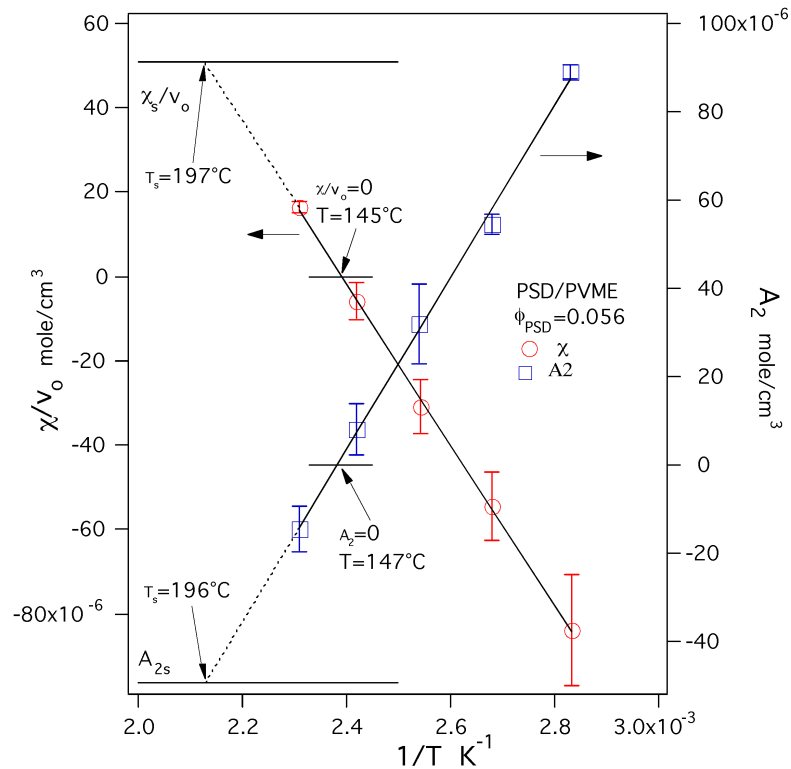
Fitting parameters:

$\bar{l}$  average segment length  
 $\chi$  Flory interaction parameter

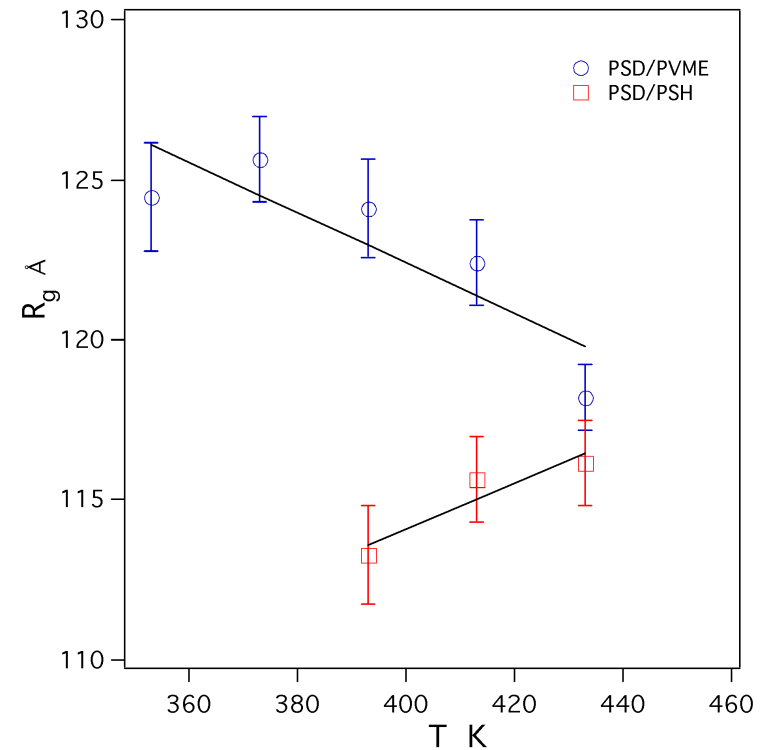
# Examples - Polymer Blends

## d-PS/PVME Blend

$A_2$  and  $\chi$  vs.  $T$



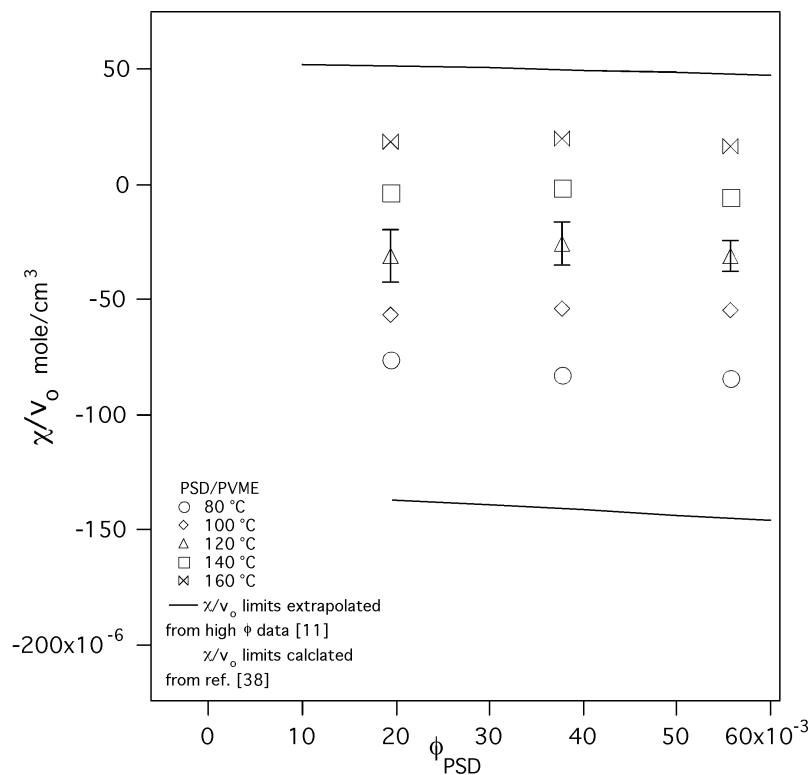
$R_g$  vs.  $T$





# Examples - Polymer Blends

## $\chi$ vs. $\phi$



Comparison of SANS data for  $\chi$  for dilute blends with:

- data extrapolated from separate study at higher concentrations
- calculated values of  $\chi$  from modified F-H lattice model

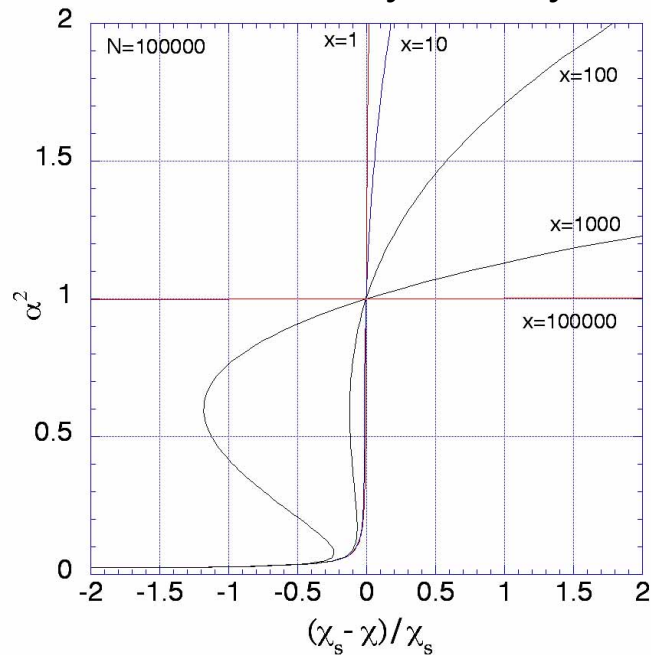
Han, C.C., et al.; *Polymer*, **1988**, 29, 2002

Dudowicz, J.; Freed, K.; *J. Chem. Phys.*, **1992**, 96, 1644, 9147

# Examples - Polymer Blends

## Collapse Transition in Dilute Blends

Prediction by Theory

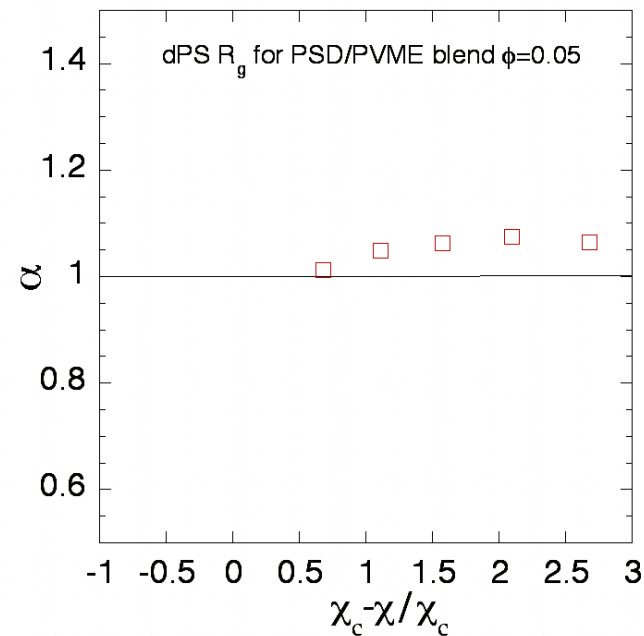


$$\alpha^2 = \frac{(R_g)^2}{(R_g^o)^2}$$

N: dilute species  
x: matrix species

Theory shows change in transition from 2nd order → 1st order with increasing matrix mol wt. (x)

d-PS/PVME data



Interesting range of prediction not accessed experimentally

# Summary



## SANS of Polymers

- **q range accessible in SANS matches intrinsic size scale of polymer ( $R_g$ ) which controls many structural and physical properties**
- **Examples of chain conformation (thin films, arborescent graft polymers)**
- **Examples of thermodynamics (blends, block copolymers)**
- **Comparison of experiments with theory (blends, block copolymers)**